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,. THE THERMAL DECOMPOSITION OF **VERMICULIT**

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Vermiculite shows some promising potentials, both as ion-exchange material and thermal insulation. The thermal dcrompositioc of vermiculite from SokJi, Finland, has been studied by various thermoanalytical methods, including high**temperature X-ray diffraction and high-pressure DTA. The major phases occurring** during successive heating are: $15\text{\AA} \rightarrow 11.8\text{\AA} \rightarrow 10.1\text{\AA} \rightarrow 9.7\text{\AA} \rightarrow 9.3\text{\AA}$ (all $d_{0.02}$) **spacings). Reaction enthalpies and activation energies are given for most of the reactions. For the evaluation of kinetic parameters, high-temperature X-ray diffraction is suggested as a powerful tool, superior to most conventional methods**

INTRODUCTION

At the Sokli carbonatite complex in Northern Finland'. *, vermiculite occurs both in phosphate rock regoliths and weathered crusts of carbonatite.

To date, the composition of the vermiculite has only been determined by energy dispersive analysis (Geological Survey of Sweden). Of the exchangeable cations, calcium seems to **predominate over magnesium; cations in octahedral coordination comprise Mg, Fe, A! and** *some Ii.*

The scope of this study was not only to depict the thermal decomposition of **vermiculite from Sokli, adding vermiculite from one more locality to those already investigated by many authors. The intention was also to use a variety of different** thermoanalytical methods - some of which have not been employed previously in vermiculite studies - and to interpret the results in terms of reaction kinetics.

EXPERIMENTAL

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For the present study, three specimens have been used:

- (1) a large single flake of vermiculite $(60 \times 50 \times 20 \text{ mm})$
- (2) as above $(40 \times 40 \times 15 \text{ mm})$

(3) a **composite sample of five large flakes.**

Specimens Nos I and 2 have been filed with a steel file, following the recommendation by Mackenzie3_ Specimen No. **3 was ground in a Culatti laboratory mi!l (rotating hammers in a steel cylinder)** *to less* **than 150 microns. All specimens were equilibrated under the following conditions:**

(1) storage in a desiccator char@ with silica 8~1 (relstive humidiiy approaching zero)

(2) **storage in a desiccator charged with saturated magnesium nitrate solution',** Relative humidity 55%.

(3) storage in distilled water, filtration immediately prior to analysis (relative humidity approaching $100\frac{6}{9}$.

(6) Equipment

The following instruments were used:

(I) L_in_seis L-62 DTA-equipment (Department of Geology, Stockholm)_ Ail sampfcs wcrc run in static air afmosphcre, in 0.09 cm3 alumina crucibks, using 'alumina as inert material. The thermocouple, Pt-Pt/lO%Kh, was calibrated qainst ICI-A reference materials.

(2) Stanton HT thermobalance (Department of Geology, Stockholm). Runs were performed in alumina crucibles in st+itic air atmosphere, at **heatins rates of 3 ana 6°C min- '_**

(3) Jeol DX-GO-S diffractometer with high-temperature equipment (Department of Geology, Stockholm). The specimens have been studied at scanning speeds **of 1 'C Qfl) min-** * **(static approach) and heating rats of 2.5 and 5°C min- ' (dynamic** approach). Pt-Pt/13%Rh thermocouples have been used.

(4) Netzsch dilatometer (AB Atomenergi, Studsvik).

(5) Netzch STA equipment (The University. Sonn, G-F-R.), with simultaneous registration of T, DTA, TG and DTG curves, at a heating rate of 10[°]C min⁻¹, in **Pt crucibles and argon atmosphere (static)_**

(6) Netzsch 404H high-pressure DTA (The University, Bonn, G.F.R.), using argon as pressure medium, at pressures in the range 5-470 bar.

RESULTS AND DISCUSSION

(a) Scheme of the thermal decomposition

The **thermal decomposition of vermiculite can be pictured as follows according** to various authors^{$3-6$}.

M-vermiculite has an initial basal spacing of 14.4-14.8 k depending upon the degree of hydration_ On heating, it changes to **a 13.8A phase in the range NM-150°C** and shortly afterwards to a 11.6-11.8Å phase. The two phases coexist in the range⁶ 150-270 °C. The structural changes are connected with the loss of four water molecules **not in immediate contact with the cation ("unbound wate?).**

With increazd temperature, five water molecules not in contact with the cation

in the completely filled single-sheet structure are removed. This is not accompanied by any structural change⁵, but is revealed by a strong endothermic DTA effect at about 170°C.

About $2^{1}/_{4}$ of the remaining 3 water molecules per cation are released at 270–300°C, accompanied by a structural change from 11.6 to 20.6Å (11.6 \div 9.02), according to Walker and Cole⁵. Bobrov et al.⁶, however, found a 9.9Å phase in the region 350.700°C, and a 10.5Å phase in the region 270-500°C slowly converting into the 9.9Å phase. They therefore conclude that the 10.5Å phase is "not an individual phase, but a mixture of the 11.6 and 9.9Å components; with an increase in temperature, the amount of the first component decreases" (op. cit., p. 532). Between 650 and 750°C, the remaining water is driven off, accompanied by a structural collapse to 9.02 (or 9.15) \AA .

Between 800 and 900°C, the DTA curve shows two endothermic effects which are identified with the stepwise dehydroxylation of the talc-like 9Å phase. In immediate succession, anhydrous mineral assemblages (commonly olivine plus enstatite) are formed.

A number of different temperatures have been selected for X-ray diffraction of specimen No. 3 of vermiculite. The results, Table 1, show fairly good agreement with the literature data quoted above. The larger initial spacing, 15.0Å, is explained by the predominance of calcium as exchangeable cation⁷. In order to establish the temperature ranges during which these phases are stable, the goniometer of the equipment was kept at constant 2θ , while the temperature was raised.

TABLE I

HIGH-TEMPERATURE X-RAY DIFFRACTION: SOKLI VERMICULITE, SPECIMEN NO. 3

Relative humidity at start: 55%

Note: Intensities refer to initial intensity = 100 .

The 15.0A phase was found to be stable from room temperature up to temperatures between 52'C (heating rate 2S"C min- r) **and about 7%75°C (heating rate** 5^oC min⁻¹). The heating rate employed is thus of importance for the stability range of the 15Å phase, as can be expected. The phase transformation $15\text{\AA} \rightarrow 13.8\text{\AA} \rightarrow$ 1 **I.sA is a consecutive reaction; consequently the existence of the I3.8A phase is not indicated by "static" high-temperature X-rdy diffraction. Although we have not studied short-lived phases such as the 13.8A phase, indirect proof of its existence is found in Fig. 1, which shows the relative abundances of the various phases. In the DTA and TG methods, this consecutive reaction is displayed as though it were a single-step first-order reaction (Fig. 2). Therefore, a strong endothermic effect is** registered on the DTA trace, with $T_1 = 55^{\circ}C$, $T_m = 144^{\circ}C$ (at a heating rate of 10°C min⁻¹). The weight loss recorded by TG is 11.9% (Fig. 2). As the weight loss during **this first reaction is desndent on the pretreatment of the sample, specimen No. 1 has been pretreated under various conditions oFhumidity. The weight losses recorded** arc (Stanton HT balance, 6^cC min⁻¹): 9.8% at 0% R.H. (relative humidity), 10.4% **at 55% R.H. and 36.5% at 10% R.H.**

The first appearance of the 11.8A phase was registered at 45"C_ The maximum amounts of the phases were recorded in the range 90-16O"C, and above 195" the phase was no longer present (Fig. I). Walker and Cole' give a stability range of 160-27O'C, and Bobrov et aL6 a range of 150-350°C for the simple 11.8A phase (up to 50O'C for the mixture of 1 I.8 and 9.9A phases). Again, the breakdown of the I1.8Å phase proceeds as a consecutive reaction $11.8\text{\AA} \rightarrow (20.6 \text{ or } 10.5\text{\AA}) \rightarrow 9.9-10.1\text{\AA}.$ **In this case also, the reaction appears on DTA and TG as though it were a single-step reaction (Fig. 2): on the DTA, a strong endothermic effect with** $T_m = 256^{\circ}\text{C}$ **is** recorded, accompanied by a weight loss of 4.0%. Data on the kinetics of the reaction.

Fig. 1. Scheme of the phases investigated, according to high-temperature X-ray diffraction. Sokli vermiculite, specimen No. 3. Heating rate, 5°C min⁻¹.

Fig. 2. Simultaneous DTA-TG-DTG of Sokli vermiculite, specimen No. 3, Netzsch STA. 125 mg sample against 125 mg alumina; Pt-crucibles, static argon atmosphere, Pt-Pt/10%Rh thermocouples, heating rate 10 °C min' ¹. Height of figure corresponds to 0.05 mV on AT , 12.5 mg on TG and 0.1 mV == 1.25 mg min^{-1} on DTG.

both stcpwise and as a whole, will be discussed below.

The 10.1Å phase forms at about 160° and starts to decompose at about 560°C ; the decomposition is completed at about 700 $^{\circ}$ C. Corresponding values are 270-650 $^{\circ}$ C (ref. 5) and $300-700$ °C (ref. 6). Our present investigation agrees most closely with the one by Bobrov et al., especially with regard to the d-spacing. Our data *indicate that* a slight but continuous loss of weight takes place within the stability range of this phase **(Fig_ 3), which becomes more marked above 550°C and is there accompanied** by an endothermic effect on the DTA (Fig. 2). The breakdown of the 10.1\AA phase is revealed by DTA as an endothermic effect, with $T_i = 646^{\circ}$ C, $T_m = 752^{\circ}$ C (Fig. 2). The weight loss during this reaction amounts to 1.7%.

The 9.7A phase appears shortly after 500°C. and is decomposed at 93O'C. However, within this range, the **9_7A phase seems to contract to a 9.3A phase. This** transition seems to proceed continuously, according to high-temperature X-ray diffractograms taken in the "static" manner (Table 1). The "dynamic" approach, however, reveals that the 9.7Å phase starts to decompose at about 710 \degree C, but is present in steadily decreasing amounts up to 93O'C (Fig. 1). The **9.3A** phase starts to decompose at 850°C, and is finally decomposed at 930°C. The transition $9.7\text{\AA} \rightarrow$ 9.3Å is revealed by DTA as an endothermic effect preceding the final decomposition and by TG as **a loss of weight** occurring in two steps. This is most clearly demonstrated by the DTG curve (Fig. 2). The weight losses recorded by TG are 0.6% for $9.7\text{\AA} \rightarrow$ 9.3Å, and 2.7% for $9.3\text{\AA} \rightarrow$ anhydrous.

The final decomposition of vermiculite as a three-step reaction has not been recorded previously. However, the evidence presented here seems conclusive.

Fig. 3. Dilatometric curves for Sok!i vermiculites.

It should also be mentioned that, of all the reactions which occur, only the first one $(15\text{Å} \rightarrow 13.8\text{Å} \rightarrow 11.8\text{Å})$ is reversible. After heating to 80^oC (complete **breakdown of the 15Å phase), a sample was allowed to cool in the diffractometer. Mter about 30 min cooling, the vermiculite had recovered to such an extent that the 15A** *peak was clearly* **visible. After an additional 30 min, the intensity of the ISA peak had reached a level of 30% of the initial intensity.**

Sawn pieces of Sokli vermiculite (specimens Nos 1 and 2) have been investi_gated with the dilatometer. A heating rate of 5^cC min⁻¹ in air was employed. All length **changezr refer to** chanses paraliel to **the crystaJlo_mphic c axis of the crystals.**

IBuring heating. all specimens show an initial contraclion, amounting from -5 *to* -9% (Fig. 3). A more or less rapid expansion follows, amounting from $\div 17$ to $\div 63\%$ in the range 150–200°C. Subsequently, a contraction is noted in the range 250–300[°]C, followed by an increase in length up to 400[°]C. It is difficult to **compare dilatometric curves with the results obtained by other methods. It seems evident, however, that the formation of the 11.8A phase causes a contraction of the material, while the formation of the 10.iA phase causes a pronounced expansion.**

(b) Es-ahiafion **of** *kinetic* pzrumctcrs

From the DTA trace, Fig. 2, the following values for *n* have been calculated **(Table 2) according to the method by Kissinger⁹:**

It can thus be stated that the (overalJ) reaction order of all reactions occurring is $n - 1$, as is expected from the reaction types involved (simplified: $A \rightarrow B + C$).

In addition, reaction orders have been determined by the method of Cyulai and Greenhow⁸, using the TG-DTG traces (Fig. 2). For the reaction $9.7\text{\AA} \rightarrow 9.3\text{\AA}$, *n* is unity. In all other cases, *n* was found to be in the range 1/3-4/5, indicating that

176

the method is not very useful. Activation energies have been calculated in a number of ways:

(I) From DTA-traces according to Kissinger⁹, using heating rates in the range $2-50$ ^cC min⁻¹ (5 points).

(2) From DTA-traces according to Piloyan¹⁰, using 5-7 points each, at a **Seating rate of IO'C min- I.**

(3) From TG-traces according to Coats and Redfern", using 5 points each, at a heating rate **of 10°C min- '.**

(4) From high-temperature X-ray dikactograms, heating rate 5°C min- ', **by the method outlined in the following.**

Starting from the rate equation $dx/dt - k(1 - \alpha)^n$, where x denotes the fraction of A decomposed at time *t* for the reaction $A_{(1)} \rightarrow B_{(1)} + C_{(2)}$, Coats and Redfem¹¹ devised a method for the calculation of E from TG traces. For first order reactions, they arrived at the conclusion that a plot of log $[-\log(1-\alpha)] \cdot T^{-2}$ against $1/T$ will result in a straight line with the slope $-E/2.3R$. Since the intensity recorded on the **difiracto_grams rclkcts the amounts of phase present at a certain time, the method of Coats and Redfern is directly applicable to dynamic high-temperature X-ray difir-ction, if the decomposition of a phase is registered. As a matter of iact, this represents a more direct approach than the &se of TG traces (where the amount of A** is **determined** by the amount of C lost). In the case of the formation of a phase, the rate equation can be rewritten as $-d\beta/dt = k\beta^2$, and the Coats-Redfern plot is thus $\log(-\log \beta/T^2)$ versus $1/T$, where β denotes the fraction of B formed at time *t*.

We are aware that a number of difficulties exist when applying this method:

(1) tinderestimation of amounts may occur, owing to the detection limit of the apparatus.

(2) Imperfect orientation of the minerals also causes underestimates.

(3) Mineral orientation may change during the reaction, either stepwise or continuously.

Several of these, and other, difficulties could be overcome by careful calibration **work, such as mixing the material with non-reacting material, However, the intention of this paper has not been to present a thorough account of the use of high-temperature X-ray diffraction for the estimation of kinetic parameters. Much more work is required,** and we feel that the promising results obtained in this study justify further investiga**tious of this topic.**

TABLE 3

ACTIVATION ENERGIES (kcal mol⁻¹) OBTAINED BY VARIOUS METHODS OF CALCULATION

Note: The first two reactions cannot be separated by DTA and TG; "bulk values" are given.

The results of all calculations performed have been compiled in Table 3. For the first reaction, no calculations based on X-ray data have been performed, owing to the rather poor temperature control of the apparatus at temperatures below 100° C. The methods based on DTA and TG yield values for the activation energy in the range 8.5 to 14 kcal mol^{-1} . For the second reaction, conventional methods of calculation show values of 15.5, 28.2 and 32.2 kcal mol^{-1} , respectively (Table 3). Calculations based on X-ray data give values of about 51 kcal mol⁻¹ for the first, and 80 kcal mol⁻¹ for the second of the two consecutive reactions. In this case, we regard the values arrived at from X-ray data as being more reliable. The method of Kissinger gives much too low a value, even when compared with the results obtained by the method of Piloyan and Coats and Redfern (Table 3). The discrepancy between these values and the ones obtained by X-ray data is readily explained. The reaction is not a simple first-order reaction, as expected from the DTA and TG traces. Instead, it is a consecutive reaction, made up of two first-order reactions. Two reactions with fairly slow reaction rates superimposed on each other yield a picture of one reaction, with a faster reaction rate and thus a lower activation energy. Considering rate equations for simple first-order reactions and consecutive reaction¹², respectively, it is obvious that the results obtained by treating the reaction as an overall first-order simple reaction must be erroneous.

For the third reaction, the value of the activation energy of 30 kcal mol⁻¹ calculated from X-ray data falls between the data derived from DTA and TG (22 and 43 kcal mol⁻¹). For the fourth reaction, X-ray data and DTA show excellent agreement; TG data yield a value that is about twice as high. For the fifth reaction (final breakdown), all values are above 100 kcal mol⁻¹; X-ray data indicate 171, DTA 101, TG 128 kcal mol⁻¹. To summarize, it can be stated that estimates of the activation energy seem to be as reliable (or unreliable) as all other conventional methods. However, the method has the advantage that a given reaction can be pin-pointed from both the decomposing and the forming constituent. Furthermore, a detailed study of

178

consecutive reactions (as well as parallel reactions) can be made, which is not possible with conventional methods. The fact that good agreement can be achieved between **independent dynamic methods seems to be verified.**

(c) Dependence of E on pressure

Using the method of Piloyan (see above), the activation energies for the first and second reaction indicated on high-pressure DTA traces have been calculated. For the first reaction (15Å \rightarrow 13.8Å \rightarrow 11.8Å), the activation energy decreases with **pressure (Fig. 4), which can be expressed in the equation** $E = 84.117 - 13.390 \ln P$ **(coefkient of correlation 0.99).**

As the reaction involves the release of a gas phase (water), it should be inhibited by increasing pressure. But, as shown above, the reaction proceeds with a decrease in volume. Thus, the reaction follows Le Chatelier's principle, and proceeds faster the **higher the pressure is.**

For the second reaction (11.8 \rightarrow 10.5 \rightarrow 10.1Å), for pressures of 5 and 100 bars a similar trend is indicated. However, with increasing pressure ($100 \rightarrow 165 \rightarrow 380$ **bars) a contradicting trend is noted: theactivation energy rises with increasing pressure_**

Fig. 4. High-pressure DTA of vermiculite specimen no 3: E versus lnP.

Fig. 5. High-pressure DTA of Sokli vermiculite, specimen No. 3, P-T plot.

As this reaction involves expansion of the material, this behaviour can be understood. At present, it is not clear why the reaction at 5 bars pressure deviates from this trend; a different reaction mechanism might be involved.

(d) Enthalpy (AH)

The enthalpies of the first two reactions registered by DTA were calculated from high-pressure DTA data. High-pressure DTA was carried out in an argon atmosphere, principally because H_2O pressure, which would have been most realistic from a geological point of view, was not obtainable.

A plot of T_i versus pressure, Fig. 5, shows that the high-temperature phases have higher molar enthalpy and higher molar volume than the low-temperature phases. The value for $\Delta V_{\text{(mol)}}$ decreases steadily with increasing pressure, until it approaches zero at pressures above 300 bars, and temperatures in the range 300–400 °C. This is explained by water being in the supercritical state under these conditions (the value of $AV_{(mod)}$ is approximately equal to the amount of water released).

180

Fig. 6. High-pressure DTA: $\ln P$ versus $1/T$. Slope = AH/R .

The enthalphy of the reactions can be determined by using the Clausius-Clapeyron equation

$$
\frac{\mathrm{d}P}{\mathrm{d}T} = \frac{\Delta HP}{RT^2}
$$

or **after integration,**

$$
\ln P = \frac{\Delta H}{RT} + C
$$

Thus, a plot of $\ln P$ against $1/T$ will result in a straight line with the slope $\Delta H/R$ The plot, Fig. 6, yields a straight line, even if there are rather few data on the low**pressure side.**

Using linear regression analysis, the following values for AH have been **calculated:**

 $15\text{\AA} \rightarrow 13.8\text{\AA} \rightarrow 11.8\text{\AA}$, $\Delta H = 21.3$ kcal mol⁻¹ (coeff. of correl. 0.98). **11.8Å** → **10.5Å** → **10.1Å**, $\Delta H = 21.2$ kcal mol⁻¹ (coeff. of correl. 0.96).

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